

Superstructure of α -phase potassium nitrate

Gunadi Adiwidjaja and Dieter Pohl*

Mineralogisch-Petrographisches Institut, Universität Hamburg, Grindelallee 48,
D-20146 Hamburg, Germany

Correspondence e-mail: mi2a002@mineralogie.uni-hamburg.de

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The structure of potassium nitrate, KNO_3 , has been redetermined at room temperature. The compound surprisingly shows a $2 \times 2 \times 1$ superstructure and crystallizes in space group $Cmc2_1$. This result contrasts with that found in former investigations, which gave the supergroup $Pm\bar{c}n$, neglecting the superstructure. The improved results are due to the employment of a CCD area detector.

Comment

Potassium nitrate is known to be both a mineral (niter) and an inexpensive chemical compound. On heating, it undergoes two phase transitions at atmospheric pressure, the first from the α phase to the β phase and the second from the β phase to the melt. These transitions are widely used to calibrate differential thermal analysis (DTA) devices (Barshad, 1952; Greis *et al.*, 1985). The crystal structure of the α phase was determined by Edwards (1931), and redetermined by Nimmo & Lucas (1973) and Holden & Dickinson (1975) using neutrons and X-rays, respectively. All these authors found an aragonite-type structure, with space group $Pm\bar{m}c$.

For many years, we have used potassium nitrate to calibrate a DTA apparatus and to teach graduate students phase transitions observed by DTA and X-ray powder diffraction. Then, in addition, we advantageously used the same compound to teach structure determination. To study the structure, we employed a Weissenberg camera and an Enraf-Nonius CAD-4 diffractometer. The Weissenberg photographs never showed any sign of superstructure, and therefore the known space group, $Pm\bar{m}c$, was confirmed. However, the situation was completely different when we used a Nonius KappaCCD diffractometer. In the latter case, space group $Cmc2_1$ was determined, with doubled crystallographic a_0 and b_0 axes. All corresponding reflections with odd h or k indices have weak intensities, thus showing a $2 \times 2 \times 1$ superstructure. On the basis of the criterion $I > 2\sigma(I)$, 1069 reflections are classified as significant, of which 1012 belong to the sublattice and 48 to

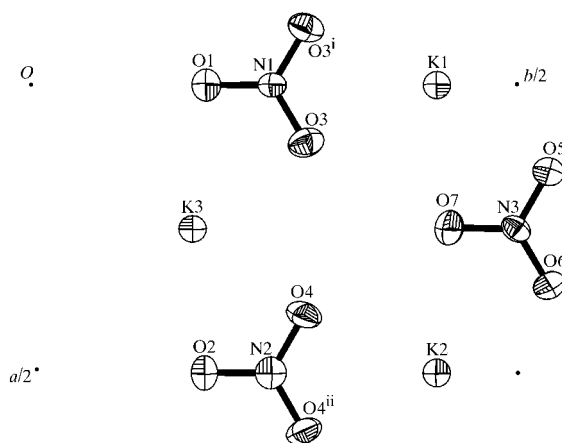


Figure 1

A view along $[001]$ of the α - KNO_3 superstructure, showing an asymmetric unit with N–O bonds. The corners of the cell are indicated with dots. Displacement ellipsoids are drawn at the 50% probability level. [Symmetry codes: (i) $-x, y, z$; (ii) $1 - x, y, z$.]

the superlattice, including 17 Friedel pairs. The intensity of the strongest superlattice reflection is less than 2% of that of the strongest sublattice reflection.

The space group of the superstructure ($Cmc2_1$) is a subgroup of that given in the literature ($Pm\bar{c}n$) for α - KNO_3 . As can be seen from the *International Tables for X-ray Crystallography* (1983, Vol. A), both space groups are connected via the minimal supergroup (maximal subgroup) $Pmc2_1$ ($Cmc2_1 < Pmc2_1 < Pm\bar{c}n$). According to the work of Lima-de-Faria *et al.* (1990), the superstructure of α - KNO_3 and the formerly described aragonite-type structure of this compound are homeotypic. Regarding the geometric properties, the superstructure differs little from the aragonite-type structure. However, the superstructure has a lowered symmetry and hence a higher number of crystallographically independent atoms in the asymmetric unit. While the true aragonite-type structure has one unique K atom and one unique N atom, each of them sitting on a mirror plane, the superstructure contains three unique atoms of both potassium and nitrogen. Each of these elements occupies two unique sites on a mirror plane and one at a general position, as depicted in Fig. 1.

There are seven unique O atoms in the superstructure of α - KNO_3 . All seven are bonded to N atoms, forming nitrate ions, of which three are unique (see Table 1 for selected bond lengths and angles). In each nitrate ion, the N atom deviates by 2 s.u. or less from the plane through the O atoms. Each of the three unique potassium ions is surrounded by nine O atoms from six nitrate ions, thereby building an irregular coordination polyhedron.

A face-indexing analytical absorption correction was applied, based on a crystal bound by ten faces, *viz.* $(0\bar{1}\bar{1})$, $(01\bar{1})$, (010) , $(0\bar{1}0)$, (110) , $(\bar{1}\bar{1}0)$, $(0\bar{1}\bar{1})$, $(\bar{1}11)$, $(\bar{1}10)$ and $(1\bar{1}0)$, with face-to-centre distances of 0.11, 0.06, 0.16, 0.17, 0.13, 0.11, 0.12, 0.15, 0.10 and 0.17 mm, respectively. The crystal neither looks orthorhombic nor resembles a sphere, thus making absorption correction extremely difficult.

Experimental

Crystal data

KNO ₃	Mo K α radiation
$M_r = 101.11$	Cell parameters from 2170 reflections
Orthorhombic, $Cmc2_1$	$\theta = 1.0\text{--}40.3^\circ$
$a = 10.825$ (6) Å	$\mu = 1.46$ mm ⁻¹
$b = 18.351$ (10) Å	$T = 293$ (2) K
$c = 6.435$ (3) Å	Irregular, colourless
$V = 1278.3$ (12) Å ³	$0.33 \times 0.25 \times 0.17$ mm
$Z = 16$	
$D_x = 2.101$ Mg m ⁻³	

Data collection

Nonius KappaCCD diffractometer	3857 independent reflections
φ and ω scans	1069 reflections with $I > 2\sigma(I)$
Absorption correction: analytical face-indexing (COLLECT; Nonius, 1999)	$\theta_{\max} = 40.2^\circ$
$T_{\min} = 0.65$, $T_{\max} = 0.78$	$h = -19 \rightarrow 19$
3857 measured reflections	$k = -33 \rightarrow 33$
	$l = -11 \rightarrow 11$

Refinement

Refinement on F^2	$(\Delta/\sigma)_{\max} = 0.036$
$R[F^2 > 2\sigma(F^2)] = 0.037$	$\Delta\rho_{\max} = 0.46$ e Å ⁻³
$wR(F^2) = 0.160$	$\Delta\rho_{\min} = -0.35$ e Å ⁻³
$S = 1.04$	Extinction correction: SHELXL97
3857 reflections	Extinction coefficient: 0.0097 (7)
101 parameters	Absolute structure: Flack (1983),
$w = 1/[\sigma^2(F_o^2) + (0.0308P)^2 + 2.4843P]$	1632 Friedel pairs
where $P = (F_o^2 + 2F_c^2)/3$	Flack parameter = 0.1 (5)

In order to obtain the maximum number of superstructure reflections, we chose a high angle, θ_{\max} , of measured reflections. Therefore, and because of the general weakness of the superstructure reflections, we unavoidably obtained many reflections below the $2\sigma(I)$ threshold, making the refinement a little clumsy. Moreover, the pseudosymmetries imposed by the substructure result in strong correlations of several parameters. Additional correlations are possibly introduced *via* absorption-correction errors (see above). As a result of the correlations, the convergence of the refinement is poor. The use of damping parameters was necessary in the last cycles of refinement, and a final cycle with no damping was added. Because of the presence of a pseudo-centre of symmetry, the absolute structure could not be determined. Simple twinning models based on primitive monoclinic lattices neither resulted in a better convergence of the refinement nor removed the pseudo-centre of symmetry. However, if the transformation of potassium nitrate from the trigonal ferro-

Table 1

Selected geometric parameters (Å, °).

N1—O1	1.241 (12)	N2—O4	1.262 (6)
N1—O3	1.255 (6)	N3—O6	1.235 (8)
N1—O3 ⁱ	1.255 (6)	N3—O5	1.237 (8)
N2—O2	1.250 (12)	N3—O7	1.253 (9)
N2—O4 ⁱⁱ	1.262 (6)		
O1—N1—O3	120.4 (5)	O4 ⁱⁱ —N2—O4	120.4 (10)
O1—N1—O3 ⁱ	120.4 (5)	O6—N3—O5	119.8 (7)
O3—N1—O3 ⁱ	119.2 (10)	O6—N3—O7	121.5 (6)
O2—N2—O4 ⁱⁱ	119.8 (5)	O5—N3—O7	118.6 (6)
O2—N2—O4	119.8 (5)		

Symmetry codes: (i) $-x, y, z$; (ii) $1 - x, y, z$.

electric high-temperature γ phase to the room-temperature α phase is led by symmetry then, in the α phase, a C -centred cell lacking a centre of symmetry is desired. Therefore, the pseudosymmetries may be true features of the α -phase structure.

Data collection: COLLECT (Nonius, 1999); cell refinement: COLLECT; data reduction: DENZO (Otwinowski & Minor, 1997); program(s) used to solve structure: SHELXS97 (Sheldrick, 1990); program(s) used to refine structure: SHELXL97 (Sheldrick, 1997); molecular graphics: ORTEPIII (Burnett & Johnson, 1996); software used to prepare material for publication: SHELXL97.

Supplementary data for this paper are available from the IUCr electronic archives (Reference: IZ1035). Services for accessing these data are described at the back of the journal.

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